

Colloidal Hollandite Holey Rods Produced by Presynthetic Nanohybridization

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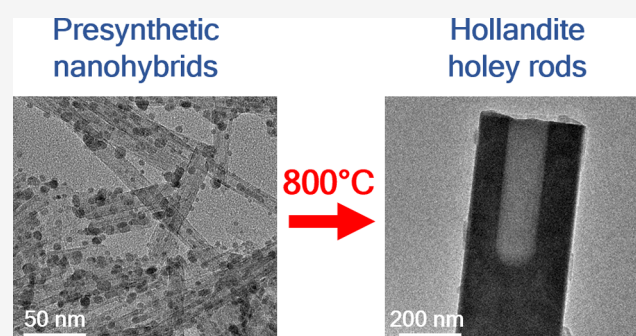
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Supporting Information

ABSTRACT: Electrostatically stabilized binary hybrids comprising TiO_2 nanotubes and Fe_2O_3 nanoparticles were self-assembled and investigated as precursors for a KFTO material. Presynthetic nanohybridization is a way to organize the components, with the caveat that the mere nanomaterial combination cannot grant a high degree of control due to their general susceptibility to aggregation, resulting in masses with poor spatial order. Various hybridization conditions were explored, and the effects of the experimental parameters were investigated in detail, considering KCl concentration, Fe/Ti ratio, and hydrothermal treatment temperature. The optimized synthetic product was obtained at a remarkably low temperature (800 °C), and it was characterized by small size, partially hollow morphology (cavity diameter ca. 100 nm), and water colloidal stability, likely inherited from the parent nanotubes. These hollow rods can be envisioned as nanoreactors for confined space synthesis and as tools for environmental remediation.

KEYWORDS: titanate nanotubes, SPIONs, nanoassembly, reagent confinement, ion exchange, nanomaterials



The opportunity to combine different nanomaterials into binary, or even more complex hybrid systems, has the potential of expanding the boundaries of nanotechnology, responding to the demand of tailoring, processing, and using functional nanostructures in critical application fields, such as nanobiomedicine,^{1–5} biosensors,^{6,7} quantum devices,⁸ and soil/water remediation.⁹ Regardless of the difficulties associated with multi-nanoparticle fabrication, the ability to custom-design these architectures would represent an advancement for their rapid in-field application.¹⁰ Among the most attractive properties of these materials there are those resulting from the collective interactions across the assembly nanocomponents and controlled by nanoparticle features, for example, secondary structure orientation phase, symmetry, or dimensions.^{11–13}

Here, nanomaterial hybridization was used as a strategy for confining reagents at the nanosize level, and the effects of this approach on the formation of hollandite were compared to the synthetic route that employs bulk iron and titanium oxides as reagents.¹⁴ Recently, hollandite supergroup minerals attracted interest for their biological inertness, chemical stability, fast ion conduction, and ion exchange ability,^{15,16} which make them suitable adsorbents for radioactive elements or solid electrolytes.^{17–21} Moreover, hollandite-based semiconductors were widely employed for electrochemical energy storage.^{22,23}

The stoichiometry of hollandite materials can be represented by the general formula $\text{A}_x\text{B}_8\text{O}_{16}$, where A is an alkaline and/or

alkaline-earth cation, while B is a metal such Al, Fe, Cr, or Ti, with varying oxidation states.²³ Potassium titanates with a hollandite structure possess a high mobility of potassium ions in the channels of titanium–oxygen octahedra, but exhibit a low electrical conductivity (down to 10^{-8} S cm^{-1}).^{24,25}

In the burgeoning context of new functional material research,^{26,27} the current study focused on developing a titanium oxide derivative comprising potassium, iron, titanium, and oxygen (KFTO), namely, a Fe(III)-doped potassium titanate commonly referred to as the hollandite supergroup.²⁸ Indeed, different protocols were proposed for the synthesis of KFTO hollandite, which produced rods with various sizes, purities, and properties.^{14,21,24,29–31} In this view, the main challenge in developing novel materials is to control the nucleation and growth stages during the reaction.³²

Herein, an electrostatically stabilized hybrid was self-assembled by a simple water incubation of oppositely charged

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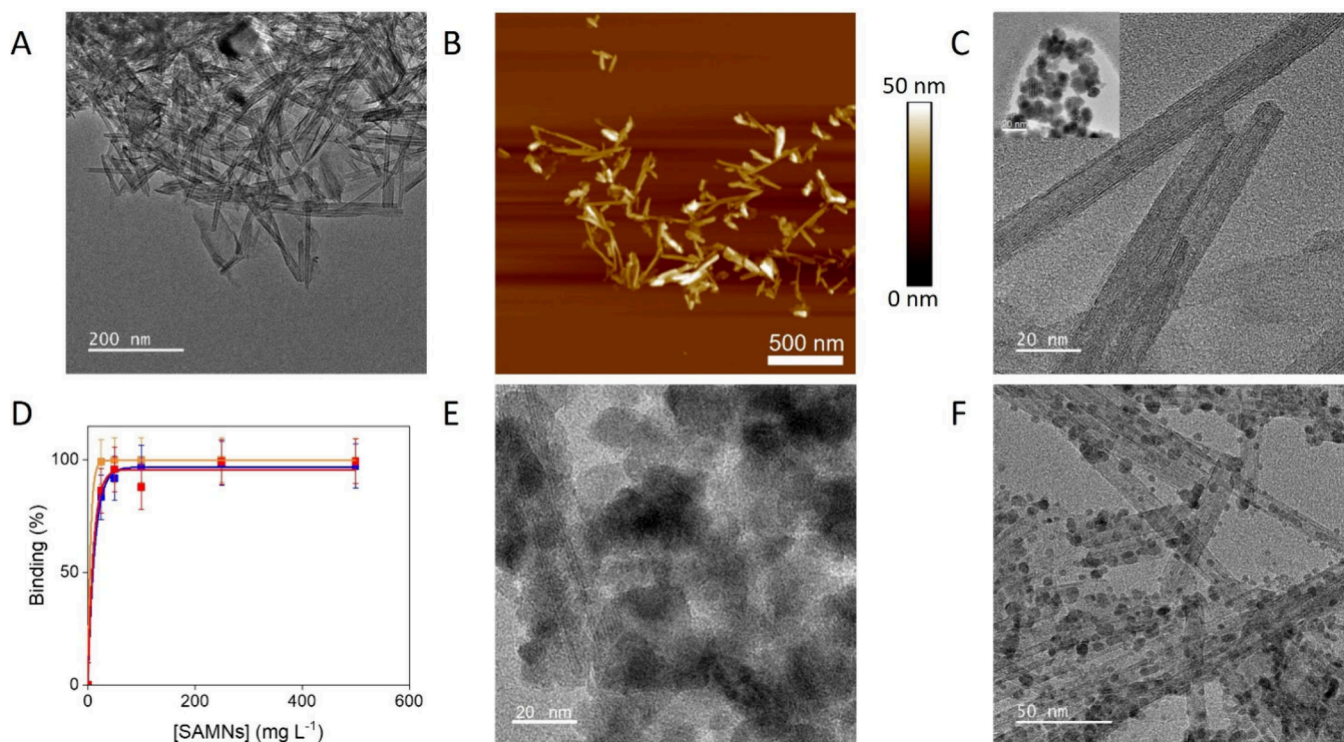


Figure 1. Self-assembly wet reaction between TiNTs and SAMNs. (A, B) TEM bright field (BF) and AFM images of parent TiNTs. (C) Higher magnification TEM-BF image showing TiNT multiwalled structure, inset: HR-TEM image of isolated SAMNs. (D) Percentage of magnetically captured TiNTs as a function of magnetic nanoparticle concentration and 10 mM (red line), 20 mM (blue line), or 50 mM (yellow line) KCl. Error bars represent standard deviations; measurements were performed in triplicate. (E) TEM-BF image of TiNTs-SAMNs hybrids obtained through the self-assembly reaction in the presence of an excess of maghemite and (F) TEM-BF image of TiNTs-SAMNs hybrids obtained from the incubation of 100 mg L^{-1} of TiNTs and 25 mg L^{-1} of SAMNs.

TiO₂ nanotubes (TiNTs) and surface active maghemite nanoparticles (SAMNs).

After its thermal treatment at $800 \text{ }^\circ\text{C}$, a novel KFTO hollandite was obtained and thoroughly characterized by Fourier-transform infrared spectroscopy (FTIR), μ -Raman spectroscopy (MRS), atomic force microscopy (AFM), transmission electron microscopy (TEM), scanning transmission electron microscopy (STEM), dynamic light scattering (DLS), and X-ray powder diffraction (XRPD), showing a hollow cylindrical morphology, submicron size, and outstanding colloidal stability in water.

The synthesis presented in this work could expand the range of hollandite technological opportunities. In fact, besides the general properties that convey interest in these materials, synthetic hollow nano- and micromaterials represent elective platforms to develop nanoreactors. The latter have an enormous potential for a wide range of applications, from environment remediation to energy storage,³³ mimicking the way nature fabricates functional compartments to perform complex biological processes. For example, compartmentalization for nanoreactor applications can be obtained by confining active chemical species into well-defined volumes through “postdecoration approaches” which consist in the internalization of a variety of metal particles (e.g., noble metal particles) in the preformed cavity while controlling their combination, number, and population density.³⁴ Among such approaches, the “ship-in-a-bottle” process is the most representative, and it involves the growth of the incorporated precursors into larger nanocrystals through reduction or assembly reactions, so that the active material is captured

inside the cavity.³⁵ The combination of colloidal stability and nanosized volumes makes the holey submicron rods, proposed in this work, good candidates for the fabrication of nano-reactors through the “ship-in-a-bottle” strategy. In fact, while their stability in water enables them to carry out reactions in an aqueous milieu, the submicrometer size allows their separation by centrifugation from nanosized colloidal particle excess. The present results are expected to stimulate the use of nanomaterial hybrids as a precursor for novel materials with emerging properties.

On this rationale, electrostatically stabilized hybrids were developed by a self-assembly reaction, simply mixing colloidal suspensions of TiNTs and SAMNs, as described in the **Materials and Methods** section (see **Supporting Information**). **Figure 1A–C** illustrates the morphology of the TiNTs and SAMNs. TiNTs suspension’s zeta potential (ζ) was $-37.9 \pm 0.4 \text{ mV}$ (conductivity = $-0.053 \text{ mS cm}^{-1}$ in water at $25 \text{ }^\circ\text{C}$), while SAMNs displayed a positive ζ above $+30 \text{ mV}$, as extensively reported elsewhere.³⁶ The TiNTs anisotropic hydrodynamic size is reported in **Figure S1A**.

The electrostatic interaction between TiNTs and maghemite nanoparticles is expected to be pivotal to the development of the KFTO hollandite and, for this reason, potassium was assumed to have the double role of hybrid formation-promoting agent as well as an elemental component of the new synthetic material.

Optimal synthetic conditions were investigated by screening KCl concentrations in the range between 1 and 50 mM and varying the TiO₂/Fe₂O₃ mass ratio (w/w) from 4 to 0.2. The binding efficiency as a function of SAMN and KCl

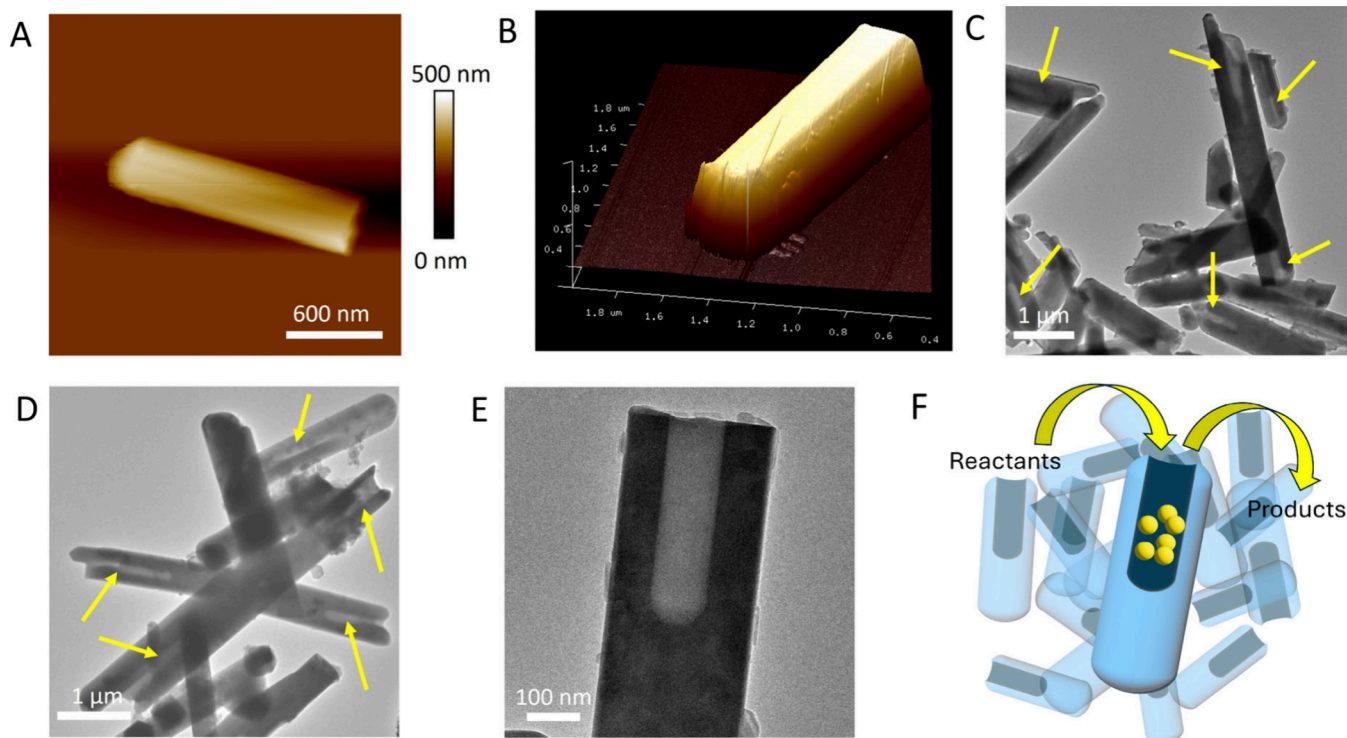


Figure 3. Morphological characterization of the synthetic products with the presynthetic hybridization step. (A and B) Atomic force micrographs showing a representative 2D and 3D image of a single tubular item of KFTO hollandite. (C and D) TEM-BF images of KFTO hollandite tubes, where the presence of cylindrical cavities in the nanosized range can be appreciated (indicated with yellow arrows). (E) TEM-BF image showing a single well-defined rod-like KFTO hollandite crystal with a sharp cylindrical cavity. (F) Schematic representation of a nanoreactor development using the preformed KFTO hollandite cavity.

precursor incubation and 600 °C curing led to a large amount of sylvite (80%). When the thermal treatment was carried out at 10 mM KCl and 800 °C curing, the scenario was still characterized by a rather heterogeneous composition that included hematite (83%), anatase (11%), rutile (2%), freudenbergite (2%), and pseudobrookite (2%), **Figure S2C**. In contrast, when the process was performed at KCl concentrations equal to 50 mM, or higher, and 800 °C curing, KFTO hollandite formation occurred (**Figure S2D and E**). Along with the desired product, hematite and sylvite were present as two main byproducts. It can be assumed that the excess of SAMNs or KCl resulted in the increased percentage of these contaminants, at the expense of the KFTO hollandite purity.

KCl concentration increment to 250 mM offered no advantages in terms of KFTO hollandite phase enrichment, while it resulted in a drastic increase of the salt byproduct (**Figure S2E**). Thus, 50 mM KCl was identified as the optimal concentration. The minor components sylvite and halite were easily and completely removed by postsynthetic washing with ultrapure water. Hematite contribution was almost completely zeroed by using 25 mg L⁻¹ SAMNs in the initial hybridization step (**Figure S2F, Table S2**). **Figure 2A** shows the trends of hematite and KFTO hollandite relative abundances as a function of the increasing TiNTs/SAMNs mass ratio. While hematite decreased, following an exponential decay, KFTO hollandite exhibited the opposite behavior and was well-fitted by an exponential growth with 100% yield at 25 mg L⁻¹ of SAMNs. The XRPD pattern (**Figure 2B**) of the KFTO hollandite phase obtained using 50 mM KCl and the 4:1 TiNTs/SAMNs mass ratio was refined through Rietveld

analysis, matching published peak positions and relative intensities.¹⁴

The results of the Rietveld refinements are presented in **Table S3**. The lattice parameters, the atomic positions, and occupancies were found to be consistent with values reported in JCPDS 98-019-2751,¹⁴ K_{1.55}(Ti_{6.5}Fe_{0.15})O₁₆, tetragonal crystal system *I4/m*; *a* = *b* = 1.01503(1) nm, *c* = 0.29717(1) nm (**Table S3**), while three views of a unit cell graphical representation of the synthetic KFTO hollandite are reported in **Figure 2C, D, and E**. FTIR and Raman spectra were collected to investigate the Fe–Ti–O bonds in the KFTO hollandite. FTIR peaks (**Figure 2F**) in the region 400–800 cm⁻¹ were ascribed to (Fe/Ti)O₆⁸⁻ octahedral modes and Ti–O–Ti vibrations.^{22,38} In particular, the signal at 780 cm⁻¹ was attributed to the stretching vibration of (Fe–Ti)O₆⁸⁻ octahedra,³⁸ while peaks at 437 and 525 cm⁻¹ were related to (Ti–O–Ti) bonds.²² Raman spectral analyses of KFTO hollandite (**Figure 2G**) displayed two sharp peaks centered at 127 and 349 cm⁻¹ and four broad features at ~273, ~491, ~603, and ~688 cm⁻¹ Raman shift. The peak at 127 cm⁻¹ was attributed to the symmetric stretching of (Fe–Ti)O₆ octahedra,³⁹ while the other above-mentioned features are the result of (Fe–Ti)O₆ octahedra bending modes.⁴⁰

As a control, the synthetic product obtained by skipping the preliminary self-assembly reaction in 50 mM KCl and the 4:1 TiNTs/SAMNs mass ratio was analyzed by XRPD after curing at 800 °C, resulting in KFTO hollandite. In the absence of the initial electrostatic driven hybridization, the cured material, when simply vortexed in an aqueous solution, showed the emergence of a macroscopic particulate. This coarse component was resistant to a prolonged ultrasound treatment

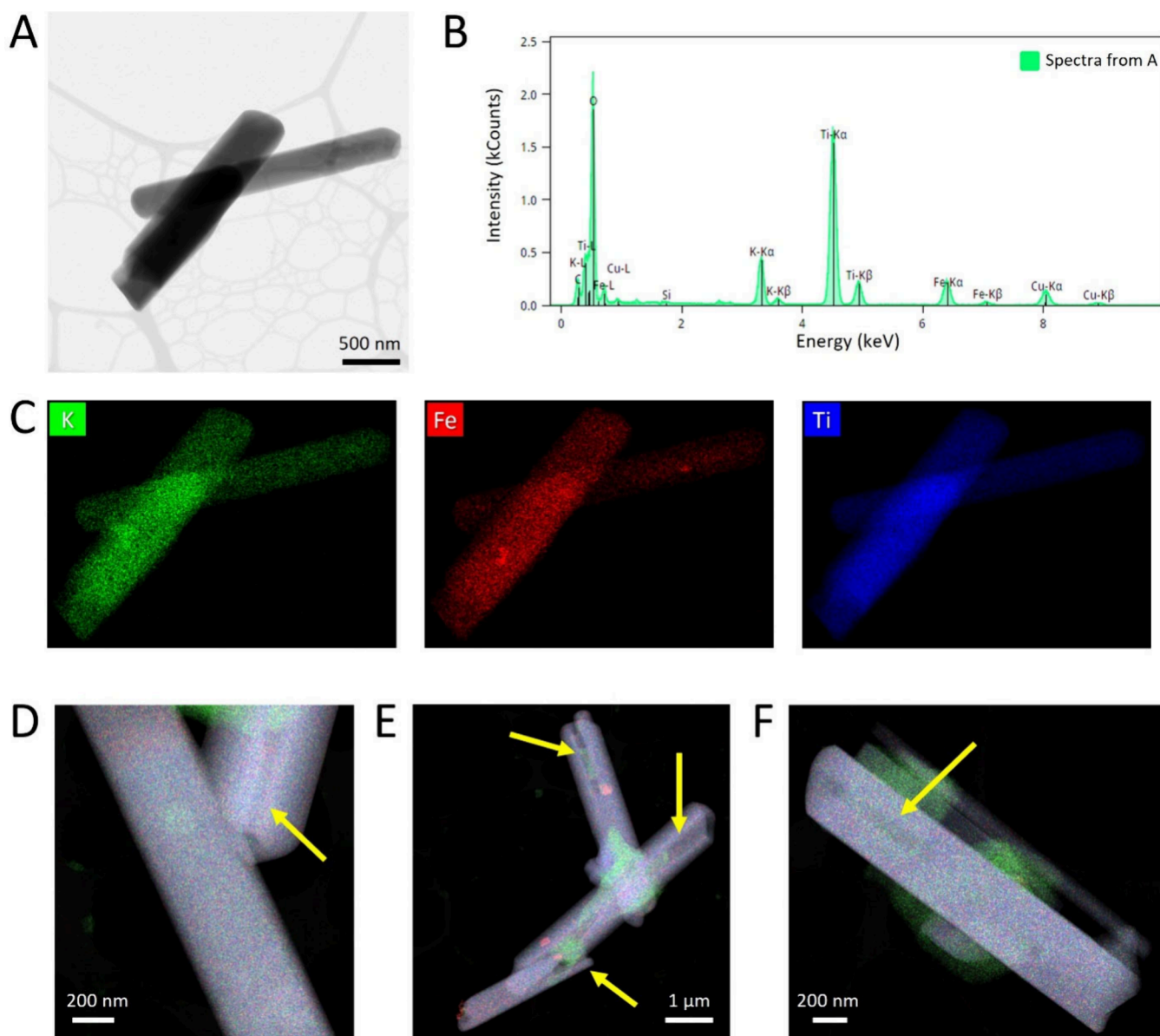


Figure 4. STEM-BF and EDS maps of KFTO hollandite crystals obtained by curing at 800 °C; hybrids self-assembled using 50 mM KCl, 100 mg L⁻¹ of TiNTs, and 25 mg L⁻¹ of SAMNs. (A) STEM image of KFTO hollandite tubes with the EDS-related spectrum shown in (B). (C) Elemental distributions (K+Ti+Fe+O) as individual maps (D) to (F) show composite EDS maps of different areas; the presence of the aforementioned cavities in the nanosized range can be appreciated as well (indicated with yellow arrows). K, Fe, and Ti are reported in green, red, and blue, respectively.

and, therefore, not analyzable by AFM or DLS. However, the zeta potential of the material present in the as-obtained supernatant was measured and it resulted to be -20 ± 6 mV (conductivity was 0.065 mS cm^{-1}), characterizing this minor water dispersed fraction as a moderately stable suspension ($-30 < \zeta < -20$).⁴¹ The correlation curve obtained by DLS analysis of the material, without the presynthetic hybridization, showed a very high polydispersity and an increase in the mean hydrodynamic size, as indicated by the high noise over $10^4 \mu\text{s}$ (Figure S3, blue line). In Figure S4A, the overall aggregated nature of the material can be clearly appreciated by TEM micrographs, showing extended bundled masses in vast regions of the sample. These apparently anisotropic items are hardly distinguishable due to their entanglement and overall morphological disorder (Figure S4B and C). The situation is further exacerbated by the presence of a matrix, apparently

merging the undefined objects together (Figure S4D). Irregular columnar shapes were identified only in some isolated areas at the edge of the large aggregates (Figure S4E and F), possessing average length and diameter equal to $2.3 \pm 0.2 \mu\text{m}$ and $490 \pm 50 \text{ nm}$, respectively (Figure S5A and B).

Conversely, the cured self-assembled SAMN-TiNT complexes resulted in a stable suspension, showing no sign of precipitation for at least six months. Furthermore, the synthetic product was observed using AFM (Figure 3A and B), showing a well-defined cylindrical shape. Indeed, the thickness of the anisotropic material is comparable to its width, suggesting that the item possesses a circular cross section and therefore an overall tubular geometry. DLS analysis further substantiated the differences between the two materials. The mean hydrodynamic size of the product obtained from the nanohybrids was $823 \pm 2 \text{ nm}$. This is the result of the median

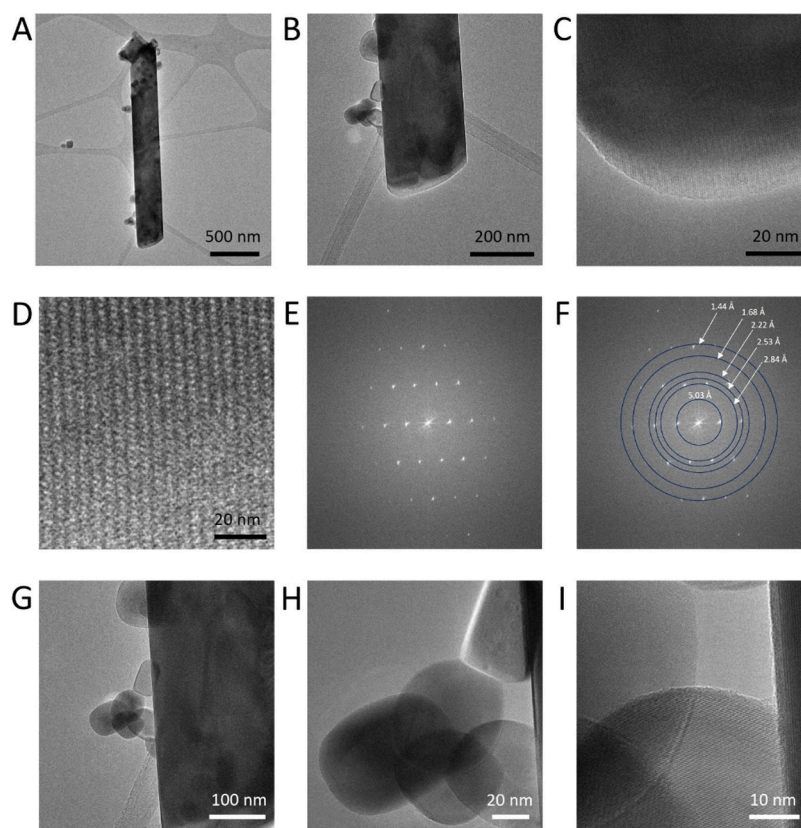


Figure 5. Maghemite nanoparticles excess influence on hollandite synthesis. In (A), the TEM-BF image of a KFTO hollandite tube surrounded by nanoparticles of hematite is shown; in (B, C), the same rod is shown at higher magnification. (D) shows a HR-TEM image of KFTO hollandite lattice fringes. In (E, F), diffractograms extracted from the HR-TEM image (C), with (F) including the d -spacing attributed to KFTO hollandite. In (G), (H), and (I), hematite nanoparticles in TEM-BF images at different magnifications.

between the length and the diameter of the anisotropic shape of the material as analyzed by AFM, which is therefore compatible with the presence of individual rods in suspension. Most importantly, colloidal stability was corroborated by a large negative zeta potential (ζ) equal to -40 ± 2 mV, with its conductivity being 60 mS cm^{-1} , representing an unprecedented feature for a hollandite material, likely a heritage of the parent TiNTs. TEM analysis further confirmed the formation of a sub-micrometric material with good monodispersity and a well-defined, partially hollow cylindrical shape (Figure 3C and D). Average length and diameter measurements were $3.3 \pm 0.4 \mu\text{m}$ and $520 \pm 50 \text{ nm}$, respectively (Figure S5C and D). Noteworthy, these sharp cylindrical cavities were clearly recognizable in many analyzed items, nicely recalling a round-bottom laboratory tube (Figure 3E). Thus, the initial self-assembly reaction is a mandatory prerequisite to obtain such a peculiar material, endowed with an unprecedented size, morphological feature, and exceptional colloidal stability. In particular, the presence of cavities is expected to hold value in the context of the ever-developing field of hollow and core-shell nanomaterials⁴² and nanoreactors,^{43,44} where nanoconfined spaces are used to obtain products with specific dimensions,⁴⁵ discover new materials,⁴⁶ build nanosensors,⁴⁷ or obtain highly efficient or highly selective reactivities.^{48–52} Figure 3F shows a scheme of the hollandite cavity as a nanoreactor through the confinement of active nanomaterials.

An EDS-equipped STEM was used to analyze qualitatively the KFTO hollandite composition, and a representative object is reported in Figure 4A, while its corresponding EDS

spectrum is shown in Figure 4B. Color-coded element representations of the same object shown in Figure 4A are reported in Figure 4C, demonstrating a homogeneous distribution of K, Fe, and Ti. In particular, the following KFTO average atomic composition fractions were calculated from the areas in Figure 4D, E, and F: K $6.6 \pm 0.7\%$, Fe $5.5 \pm 0.9\%$, Ti $26 \pm 2\%$, and O $62 \pm 2\%$ (Table S4). Additional signals were collected from different sites of the sample, showing good agreement with expected hollandite supergroup chemistry.²⁸ This further confirms the stoichiometry assessed by the XRPD analysis and, therefore, the proposed tetragonal symmetry, as previously described.²³ In Figure 4D, E, and F, it is also possible to appreciate the rod cavities in different analysed items, as indicated by the yellow arrows.

Figure 5 shows representative TEM bright images and diffractograms of the KFTO hollandite synthesized with an excess of maghemite, where it is possible to discriminate between KFTO hollandite and hematite crystals. Indeed, the sample is characterized by two morphologically distinct components, the aforementioned tubular KFTO hollandite structures and globular hematite nanoparticles (Figure 5A and B). The bottom section of the rod and its lattice fringes are visible in higher magnification HR-TEM images (Figure 5C and D, respectively). Figure 5E and F show the resulting diffractograms from the HR-TEM image, very close to a zone axis (for the zone axis determination method, see Supporting Information, Figures S2 and S6 and accompanying description). The d -spacing values reported in Figure 5F of 5.03 Å, 2.84 Å, 2.53 Å, 2.22 Å, 1.68 Å, and 1.44 Å are consistent with

the presence of a hollandite structure *sensu*.²⁸ Hematite globular nanoparticles appear to be attached to the rod surface in a fashion similar to the native SAMNs-TiNTs complex, to some extent (Figure S5G, H, and I). This secondary nanosized material, displaying a diameter of approximately 50 nm, is attributed to the hematite as a contribution recorded by XRPD analysis (*vide supra*).

In Figure S7, EDS maps of potassium, titanium, and iron are consistent with hematite as the byproduct. In particular, hematite is distinct from the titanium-based oxide in terms of its higher Fe content.

Alternatively, hematite was present in the form of larger nanoparticle aggregates adhering to KFTO hollandite rods, with the latter acting as nucleation sites for globular nanoparticles (Figure S7). EDS results showed the presence of K, Ti, Fe, and O as elemental components of the tubes, clearly ascribable to KFTO hollandite, whereas the chemical composition of the nanoparticle aggregates essentially consists of Fe and O, in a ca. 2:3 elemental ratio, attributed to the Fe₂O₃ counterpart (Table S5). Cu and C signals in the EDS spectrum come from the grid used to support the sample and the TEM holder. The tendency of hematite to nucleate on the surface of the KFTO hollandite rods is unavoidable. In any case, hematite only mildly affected the purity of the optimized synthetic product.

Concluding, a synthetic strategy was developed for the production of a hollandite material. The protocol consisted of a simple hydrothermal treatment of electrostatically stabilized binary hybrids comprising TiNTs and SAMNs. Optimized conditions involved a preliminary 2 h incubation step with 100 mg L⁻¹ of TiNTs, 25 mg L⁻¹ of SAMNs, and 50 mM KCl, followed by curing at 800 °C. A new single-crystalline KFTO hollandite was obtained, characterized by a sub-micrometer tube-like shape and the presence of nanosized cavities in both ends of their anisotropic structure. Such objects showed a large negative zeta potential (i.e., -40 mV) leading to stable colloidal suspensions. When compared to similar hydrothermally generated materials, the evolution into hollandite took place at a substantially lower temperature. The present study paves the way to a new generation of synthetic materials, exploiting nanomaterial hybrids as reagents.

■ ASSOCIATED CONTENT

SI Supporting Information

The Supporting Information is available free of charge at <https://pubs.acs.org/doi/10.1021/acs.nanolett.5c01451>.

All materials and instruments used in this study, synthesis details, additional characterizations, and supplementary figures (PDF)

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Author Contributions

I.M.D. and G.R. contributed equally to this work as co-first authors. The project was conceived and coordinated by G.S., M.M., and F.V. Sample preparation, measurements, and analyses were carried out by I.M.D., G.R., S.M., A.B., Al.Cec., Au. Cen., F.T., M.B., M.F., S.A.S., and A.Bas. under the guidance of G.S., M.M., and F.E.B. while F.V., M.M., I.M.D., G.R. and S.M. wrote the first draft of the manuscript. All authors participated in the discussion, data analysis and interpretation, and revision of the manuscript.

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Notes

The authors declare no competing financial interest.

ABBREVIATIONS

TiNTs, titanate nanotubes; SAMNs, surface active maghemite nanoparticles; FTIR, Fourier-transform infrared spectroscopy; MRS, μ -Raman spectroscopy; AFM, atomic force microscopy; TEM, transmission electron microscopy; STEM, scanning electron transmission, microscopy; DLS, dynamic light scattering; XRPD, X-ray powder diffraction; KFTO, potassium, iron, titanium, oxygen; HR-TEM, high-resolution TEM; TEM-BF, TEM bright field; EDS, energy dispersive X-ray spectroscopy; JCPDS, Joint Committee on Powder Diffraction Standards

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